Electrochemical oxygen sensors for on-line monitoring in lead–bismuth alloys: status of development

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Abstract

This paper presents the state of development of oxygen sensors based on the electromotive force (emf) measurement at null current, using yttria stabilized zirconia as solid electrolyte for application in liquid lead–bismuth eutectic (LBE), which is envisaged as a nuclear coolant or as a spallation target in accelerator driven system (ADS) for nuclear waste transmutation. The assembly procedure, the calibration method, as well as the summary of the various validation tests undergone in both static and loop facilities are presented so as to define a real state of achievement and the basics needs for further studies. The sensors are efficient, accurate, rapid and reliable for research loops. However, the poor mechanical resistance as well as the effect of traces of impurities, promoting an increasing time-drift under certain conditions, are to be further studied to improve the sensor reliability for a nuclear use. The oxygen and chromium solubilities were reassessed in the process of the sensor testing, those relations are also given and discussed.
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1. Introduction

The accurate measurement of the oxygen concentration in liquid lead–bismuth eutectic, as well as in pure lead alloys for use as a coolant in nuclear systems or as a liquid spallation target for the high neutron source of an ADS, which is foreseen for nuclear waste transmutation [1,2], is a critical issue to define the active chemistry control and monitoring systems. The contamination of the liquid system by lead and bismuth oxides must be controlled for reliable and safe operation, as this can affect the ability of the fluid to flow and to extract the heat by clogging in the pipes or by depositing on the heat exchanger surface. The second requirement is to ensure an efficient corrosion protection of the structure if the oxide layer self-healing method is chosen, as well as to control the distribution of the activated products (fission, spallation, corrosion) within the liquid system, where oxygen is playing a key role.

The use of the ionic conduction properties of some solid electrolytes, and in particular the zirconia based ceramic, allows making an electrochemical cell assembly for on-line measurement of dissolved oxygen in a liquid metal system in a wide concentration range. This has been already addressed in previous papers [3,4]. The main objective was first to investigate the possibility to build an oxygen sensor for liquid lead alloy applications, and in particular to check whether the sensor outputs were reliable enough to operate the oxygen control processes. This is verified through the development of a specific calibration procedure. In addition, the second point was to assess its main operating characteristics so as to define its serviceability for pilot scale applications to real size nuclear system such as the spallation target loop or the coolant loop of a nuclear system (fast or sub-critical ADS system). It should be noticed that the Russian

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Federation paved the way to this development years ago [5,6] by developing this technology for nuclear naval propulsion, as well as the Harwell company [7] for similar liquid sodium sensors that have been used in CEA for more than 25 years on experimental loops.

2. Theoretical relation

The theoretical formulation of the sensor output, based on the Nernst relation, was previously addressed in [4], but with slightly inaccurate free energies of formation for the various oxides (reference electrode oxide, lead alloy oxides), as they were computed by the mean least squares method on the full temperature range (300 to 1300 K or more). Using the HSC chemistry database software [8], the range was downsized to a more relevant operating temperature that could be expected for a real application, 400–1000 K, and the results are presented in Table 1. The free energies are then in good agreement with the respective energies used elsewhere

using other databases ([9]: JANAF; [10]: I. BARIN 1989; [11]: the Oxide Handbook, 1982). The slight inaccuracy of these data will affect the theoretical electromotive force (emf) calculation by a few millivolts (mV) that are assumed as reasonable. Another source of scattering is due to the data coming from the lead activity relation in LBE, which affect only slightly the emf by a maximum of 10 mV, as well as from the oxygen solubility relation, which can affect the emf to a much greater extent. However, there is only one available relation for LBE, known as the Orlov relation [5], whose accuracy and measurement method are not very well known. The constants of the Nernst relations are recalculated using the latest free energies and using the same data as those described in [4] and reported in respectively, Tables 2 and 3 (theory lines) for Bi/Bi$_2$O$_3$ and In/In$_2$O$_3$ reference electrodes for a cell immersed into a LBE solution according to the following emf (E) vs temperature (T) and oxygen concentration (C_0) relationships:

$$E_{(mV)} = K_1 + K_2 \cdot T_{(K)} + K_3 \cdot T_{(K)} \cdot \ln C_{0_{(wt\ ppm)}}$$
$$\text{for } E > E^{sat}, \tag{1}$$

$$E^{SAT}_{(mV)} = K + K' \cdot T_{(K)} \quad \text{for } E = E^{sat}. \tag{2}$$

Apart from the data used for the theoretical assessment, there may be other causes for the sensor not to behave according to the theoretical relations [12,13]: the electrolyte conduction properties (slight electronic

Table 1
Main oxides free enthalpy coefficients for the 400–1000 K temperature range for one mol of O$_2$

$\Delta G^0 = \Delta H^0 - T \cdot \Delta S^0$	ΔH^0 (J/mol)	ΔS^0 (J/mol)
4/3 Bi + O$_2$ = 2/3 Bi$_2$O$_3$	−389 140	−192.6
2 Pb + O$_2$ = 2 PbO	−437 608	−199.1
4/3 In + O$_2$ = 2/3 In$_2$O$_3$	−618 674	−216.8

Table 2
Synthesis of the Bi/Bi$_2$O$_3$ sensors calibration results ($K_3 = -0.0431$ mV/K)

	K (mV)	K' (mV/K)	K_1 (mV)	K_2 (mV/K)
Theory	120	−0.054	−218	0.065
3x	127 ± 4	−0.054 ± 0.007	−343 ± 21	0.27 ± 0.03
4x	118 ± 2	−0.036 ± 0.003	−402 ± 35	0.33 ± 0.05
5x	140 ± 7	−0.07 ± 0.001	−283 ± 19	0.17 ± 0.03
6x	133.5 ± 0.2	−0.0584 ± 0.0002	−298 ± 10	0.19 ± 0.01
7x	165 ± 4	−0.117 ± 0.007	−289 ± 15	0.180 ± 0.02
Mean	137	−0.067	−323	0.23

Table 3
Synthesis of the In/In$_2$O$_3$ sensors calibration results ($K_3 = -0.0431$ mV/K)

	K (mV)	K' (mV/K)	K_1 (mV)	K_2 (mV/K)
4x	−439 ± 6	−0.04 ± 0.007	−958 ± 38	0.33 ± 0.05
5x	−394 ± 9	−0.1 ± 0.01	−920 ± 20	0.3 ± 0.03
6x	−464 ± 18	−0.016 ± 0.03	−997 ± 32	0.41 ± 0.04
7x	−351.5	−0.1672	−873.5	0.235
Mean	−412	−0.051	−937	0.32

conduction, impurities in the ceramic, etc.), the influence of the reaction at the electrode/electrolyte interface (liquid metal/zirconia reaction, or even traces of impurities depositing on the interface), the cell irreversibility (equilibrium not reached due to an oxygen transfer rate limitation especially at very low oxygen partial pressure), as well as the instrumental uncertainties. This is why a calibration method was developed.

3. Calibration, results and discussion

The procedure for the assembly of home-made sensor is now routine, as well as the procedure for the calibration based on the temperature variation when the oxygen is close to saturation. They have been described in [4]. This calibration procedure has been systematically applied to all sensors built, at least one time for each sensor, in a simple crucible operating in a common purified argon atmosphere gloves box ('JACOMEX'). Until now, up to 32 sensors were assembled following the principle given in [4]. The assembly procedure was applied 5 times since then (lot numbered $3x$ to $7x$, the first digit stands for the lot number while the x stands for the rank of the sensor in the assembly procedure): 20 Bismuth reference sensors (mp 271 °C), 12 Indium reference sensors (mp 157 °C) were built.

Although the procedure of calibration is considered to be slightly inaccurate, the voltage linear regression coefficients for both domains, either the dissolved oxygen or the saturated oxygen solution, are almost always higher than 0.9, indicating a variation according to the Nernst relations. However, other operating conditions affect the calibration result: the same procedure applied several times yields slightly different coefficients for the E vs (T, C_0) relation. For this, the only interpretation is to consider a slight irreversibility of the cell. The electrodes are not fully at thermodynamic equilibrium as the solution is static. Hence, the impurities may be dissolved or crystallized and there are probably some reactions on the electrolyte interface with the other dissolved impurities like the corrosion products (Fe, Ni, Cr) affecting eventually the electrode potential. To support this assumption, the results obtained for an inert crucible (aluminized steel by the pack cementation process) compared to those obtained with a steel crucible appear as more reliable as the linear regression coefficients are higher. This is especially true when the temperature gets below the 350 °C threshold for a steel crucible: the level of impurities is then within the same range of concentration of the dissolved oxygen (0.1 wt ppm) and the saturated line is no more linear.

This is why the coefficients of the E vs (C_0, T) relations are averaged to smooth these influences on the calibration results for a lot of sensors identically built. The

results of the calibration achieved are presented in Tables 2 and 3, respectively for the Bi and the Indium reference sensors.

In addition, the quality of the solid electrolyte itself plays a role in the results. When the yttria-doped zirconia comes from a different fabrication batch, small variations probably caused by impurities in the yttria and zirconia powders used or small variations in the fabrication procedure are affecting the calibration results. This should be further studied.

Despite this slight scattering, the agreement between the experimental sensor outputs and the theoretical predictions must be recognized as rather good. The accuracy is assessed within 5% of the voltage readings (25 mV maximum scattering for a 500 mV reading) that affect the concentration by 40% for the high concentration range to 80% for the low concentration range. The general relations, using the constant mean values given in Tables 2 and 3, are then recommended instead of the theoretical constants when calibration is not possible. Fig. 1 presents the calibration curves plotted against the theoretical curves.

Calibration is, in any case, recommended to achieve a better accuracy. However, the availability of a method to achieve a calibration on the field would be most useful to significantly increase the accuracy on the long term, as well as to regularly assess the good operation of a sensor. This could be particularly critical if the time drift that was observed on some occasion is confirmed. Indeed, other impurities may react at the solid electrolyte interface and form metal/metal oxide couples, which interfere with the electrode potential on the long term. In any case, this procedure and results have to be compared to other procedure and calibration results to better assess the accuracy of the calibration procedure as well as to understand the potential side effects.
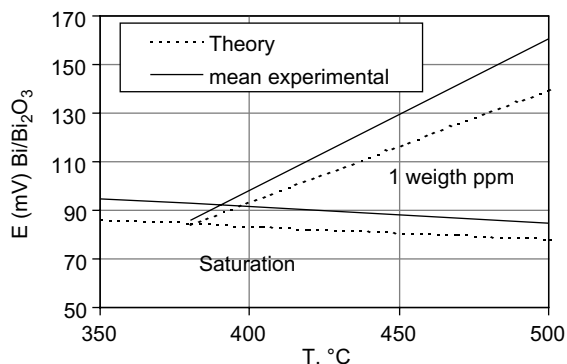

Fig. 1. Predicted Bi/Bi_2O_3 sensor outputs by using either the calibrated or the theoretical relations for the saturation line and the iso-concentration line of 1 wt ppm.

4. Oxygen and chromium solubility

It was previously stated that the only available oxygen solubility data were poorly known compared to the other required thermodynamic data. However, it is possible to use the calibration results to retrofit the oxygen solubility coefficients, A and B, as given in Eq. (3), in the relevant temperature range where the calibrations were done: 350 °C < T < 500 °C.

$$\ln C^{*}_{0_{(wt\%)}} = A - \frac{B}{T_{(K)}} \quad \text{with } A = \frac{K' - K_2}{K_3} \text{ and}$$
$$B = \frac{K_1 - K}{K_3}. \tag{3}$$

By using the following relation and the mean constants derived for each reference systems, indium and bismuth mean values given in Tables 2 and 3, it is possible to assess new solubility coefficients by meaning the two references: $A = 7.7$; $B = 11\,426$. This calculation is assuming that the mean value of both sensors, indium and bismuth, will reduce all experimental uncertainties as well as the intrinsic property variations of the solid electrolyte. When compared to the Orlov relation ($A = 2.76$; $B = 7828.8$), the solubility would be lower by half at 350 °C and about 2 times higher at 500 °C. There is however a rather good agreement with the coefficients achieved by Ghetta [15] by coulometric titration measurements in LBE ($A = 7.55$; $B = 11\,173$), whereas the coefficients calculated by Müller [14] ($A = 2.52$; $B = 4803$) are slightly lower at any temperature as presented in Fig. 2. The solubility should then be further studied as this could have an impact on the specification for a nuclear operating system as regards the coolant chemistry control.

Considering that the oxygen is at equilibrium with the saturated chromium impurity following the Cr/Cr$_2$O$_3$ equilibrium (Eq. (4)) in specific and very reductive conditions, it is possible to derive the saturated dissolved chromium concentration in the LBE solution by knowing the temperature variation of the oxygen concentration. This requires the following assumptions: the chromium activity is defined by the molar ratio, the PbO and Pb activities are defined as presented in [4],
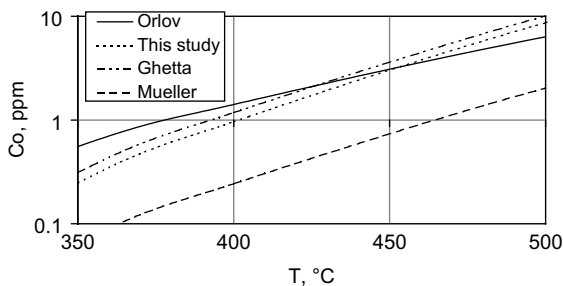
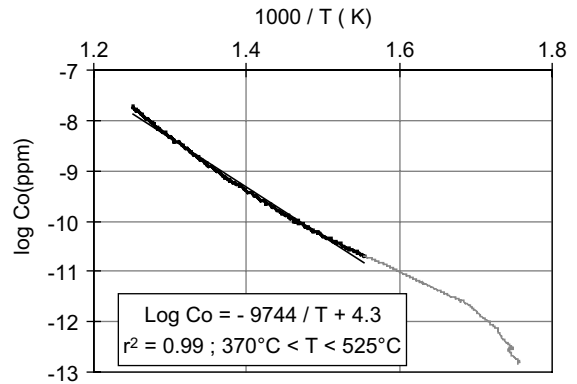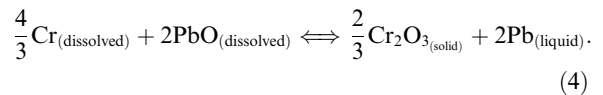Fig. 2. Comparison of the oxygen solubility relations.

Fig. 3. Oxygen measurement after hydrogen (Ar + 4% H$_2$ specially prepared) reduction at 540 °C in the BIP static facility where the chromium saturation is assumed and using the calibrated constants for the emf conversion from the oxygen sensor output (r^2 being the linear regression coefficient).

and the Orlov relation is used for the oxygen solubility. The free energy of the reaction is assessed using the data from HSC [8] in the 400–1000 K temperature range, and is given in Eq. (5).

$$\frac{4}{3}Cr_{(dissolved)} + 2PbO_{(dissolved)} \Longleftrightarrow \frac{2}{3}Cr_2O_{3_{(solid)}} + 2Pb_{(liquid)}. \tag{4}$$

$$\Delta_r G^0_{(J/mol)} = -317\,800 - 27.3 \cdot T(K). \tag{5}$$

Such a variation is presented in Fig. 3: it was obtained in the BIP static facility after hydrogen reduction. Hundreds of hours of operation of the static device, which was including some metallic parts wetted in LBE (SS304 steel), were achieved at that moment, so that the effective saturation of chromium could be assumed. Using the experimental relation presented in Fig. 3, the chromium solubility is then calculated:

$$\ln C_{Cr(wt\ ppm)} = 11.68 - \frac{6959.25}{T_{(K)}}$$
$$\text{for } 370\ °C < T < 540\ °C. \tag{6}$$

The relation is surprisingly very close to the Russian only available relation [6], whose coefficients are respectively, 9.164 and 5250. For instance, the concentrations calculated are respectively 3.9 wt ppm at 400 °C for the Russian relation and 3.8 wt ppm for this study. On that particular occasion, the lowest oxygen concentration that was ever observed in our facilities was measured: 10^{-11} wt ppm at 370 °C.

5. Results and discussion

Sensors have been tested for thousands of hours in LBE solution in both static (BIP [4], COLIMESTA

[3,4], JACOMEX [4] and wetting tests gloves boxes, COXCIMEL [15]) and dynamic conditions (CICLAD, LiSoR, and STELLA loops) in stable or cycling chemical conditions that validate their effective and reliable use although some limiting behaviors appeared. These tests allowed gaining a large operating feedback representative of long-term operations on a wide variety of operating conditions. The characteristics observed up to now on these oxygen prototype sensors are then satisfactory and confirm the previous observations [4], as regards the operating range (370–550 °C), the oxygen concentration range, the response time, the accuracy and reproducibility, as well as the service life (up to 4000 h observed, but longer expected in stable conditions). A lower operating temperature than 350 °C is possible but with a lower accuracy as the sensor outputs deviate from ideal behavior most probably because of an increasing irreversibility of the cell (Fig. 3).

Limiting factors were clearly pointed out from the sensors operations that question their reliability for long-term operation on a nuclear facility: little mechanical resistance, and the effect of the impurities on the long term for high oxygen content.

On static devices, almost no sensor breakdown was observed so long that no thermal or mechanical shocks were ensured. For instance, three zirconia thimbles were broken on the same day when skimming out superficial oxide during the calibration operation. Another sensor breakdown is illustrated on Fig. 4. In a dynamic facility, the vibrations due to turbulent flow especially enhanced by the length of the zirconia tube, led to the failure of a number of sensors, both for protected (LiSoR) or unprotected (CICLAD, four in one year) sensors. The use of a metallic sheath clearly protected the ceramic for a longer time. The local flow conditions appeared to be important, as reduced turbulence as well as lower flow velocity should be favored.

However, the use of a proper design for the ceramic housing, together with the use of special care for operation would allow reducing these causes of failure. This

Fig. 4. Details of the Bi/Bi$_2$O$_3$ reference electrode featured on a broken sensor.

was typically used for the liquid sodium sensor (Westinghouse, Harwell [7]) and is known as efficient. Although the closed-end tube shape is commercially available and easily separates the two electrodes, the working and the reference electrodes, it might be replaced by other less fragile shapes, like a conical shape or a pellet shape fused to alumina tube for instance that is reported as more resistant [13]. However, the implementation becomes more complex because of the issue of the metal-to-ceramic or ceramic-to-ceramic sealing, without speaking of the cost.

The time drift appeared as negligible with low oxygen concentration and stable operating conditions. On the other hand, high oxygen content or cycling the oxygen from low to high contents, affects the sensor output on the long term by clearly increasing the time drift, and eventually preventing any oxygen measurement. The other impurities present in the liquid metal, and especially those generated by a steel sheath close to the electrolyte, may deposit on the electrolyte interface, affecting eventually the electrode potential. The potential recovery operations, like high temperature in low oxygen washing operation, or nitric acid washing of the ceramic, should be studied as well as the basic phenomena involved.

6. Conclusion

Assembly and calibration procedures were achieved. A large testing in various conditions allowed defining a real state of achievement for the oxygen sensor. The basic static and loop implementation of the oxygen sensing ceramic proved efficient and gave reliable information for research studies. However, their uses are still not straight forward, especially for dynamic loop facilities, as a time drift apparently due to the long term influence of the impurities, as well as a fragile ceramic are observed. This makes the device not fully reliable for nuclear implementation yet, but should be solved by a proper design for the ceramic housing together with the use of special care for operation. The time drift due to the impurity effects, if confirmed, should be solved by a recovery procedure, whose possibility is to be studied as well. The time drift must be monitored on the long term, so that on-the-field calibration procedure is required.

In addition, the oxygen sensor could measure more than just oxygen. Indeed, assuming that equilibrium prevails amongst dissolved oxygen and the other solutes, it is possible to determine the concentration of other solutes from the oxygen sensor measurement, as was presented for the chromium solubility, similarly to the steel making industry that measure the dissolved carbon by this method.

Acknowledgments

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